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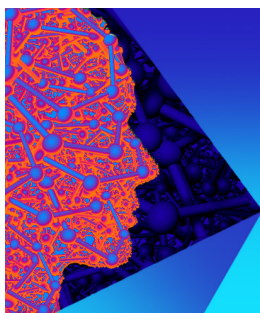


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ABSTRACT

Two-dimensional (2D) materials have gained increasing prominence not only in fundamental research but also in daily applications. However, to fully harness their potential, it is crucial to optimize their properties with an external parameter and track the electronic structure simultaneously. Magnetotransport over a wide magnetic field range is a powerful method to probe the electronic structure and, for metallic 2D materials, quantum oscillations superimposed on the transport signals encode Fermi surface parameters. In this manuscript, we utilize biaxial strain as an external tuning parameter and investigate the effects of strain on the electronic properties of two quasi-2D superconductors, MoTe₂ and RbV₃Sb₅, by measuring their magnetoresistance in pulsed magnetic fields up to 60 T. With a careful selection of insulating substrates, we demonstrate the possibility of both the compressive and tensile biaxial strains imposed on MoTe₂ and RbV₃Sb₅, respectively. For both systems, the applied strain has led to superconducting critical temperature enhancement compared to their free-standing counterparts, proving the effectiveness of this biaxial strain method at cryogenic temperatures. Clear quantum oscillations in the magnetoresistance—the Shubnikov–de Haas (SdH) effect—are obtained in both samples. In strained MoTe₂, the magnetoresistance exhibits a nearly quadratic dependence on the magnetic field and remains non-saturating even at the highest field, whereas in strained RbV₃Sb₅, two SdH frequencies showed a substantial enhancement in effective mass values, hinting at a possible enhancement of charge fluctuations. Our results demonstrate that combining biaxial strain and pulsed magnetic field paves the way for studying 2D materials under unprecedented conditions.

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Rapid progress in the discovery, synthesis, and reliable exfoliation of two-dimensional (2D) or quasi-2D materials has revolutionized our daily lives by offering an extensive range of applications. Notably, graphene-related devices have played a critical role in rechargeable battery technology, where energy efficiency has been greatly enhanced.^{1–4} The high transparency and mechanical flexibility exhibited by transition metal dichalcogenides (TMDs) have positioned them as promising candidates for next-generation solar cells.^{5–8} Furthermore, emerging fields based on 2D materials, such as “twistronics,”^{9,10} also open up new avenues for harnessing exotic

physical phenomena. Given the prominence of 2D materials in modern condensed matter research, it is necessary to thoroughly understand and control their electronic structure.

A powerful means to modify the electronic properties of bulk, three-dimensional (3D) materials is to subject them to hydrostatic pressure.^{11–13} One can then contemplate an equivalent approach for tuning 2D materials: since the hydrostatic pressure shrinks the volume of 3D materials uniformly, an effective methodology is needed to vary the area of the 2D materials uniformly. Thus, an in-plane, biaxial strain is an effective tuning

parameter for 2D materials, as has been demonstrated in Fe-based superconductors.^{14,15}

The modification of the sample volume or area changes the interatomic spacing. Consequently, the electronic structure can be altered. Indeed, not many materials come immediately with desirable properties without suitable optimizations. With the modification of the electronic structure, it is then necessary to determine the carrier densities and mobilities, and for metallic 2D materials, the Fermi surfaces.

The magnetic field is extremely useful for probing electronic properties. First, measuring electrical transport in a magnetic field enables the decoupling of the carrier mobility and density, which would otherwise be impossible to determine independently. Next, the Shubnikov–de Haas (SdH) effect, or magnetic quantum oscillations in electrical resistance, is a powerful tool for Fermi surface characterization.¹⁶ Generally, the higher the field strength, the more pronounced the quantum oscillations because the Landau levels are less broadened. Thus, the ability to tune 2D materials with biaxial strain and then conduct magnetotransport measurements at high fields will be beneficial for investigating a variety of 2D materials. In this manuscript, we demonstrate the ability to detect the SdH effect in superconducting MoTe₂ and RbV₃Sb₅ under biaxial strain in a pulsed magnet.

Single crystals of MoTe₂ and RbV₃Sb₅ were prepared by the self-flux method, as described elsewhere.^{17,18} These single crystals were used in both free-standing and biaxial strain measurements. The dimensions ($L \times W \times H$) of MoTe₂ and RbV₃Sb₅ for the biaxial strain devices are $818 \times 211 \times 14 \mu\text{m}^3$ and $804 \times 255 \times 11 \mu\text{m}^3$, respectively. The MoTe₂ sample was attached to a polycarbonate substrate with Cyanoacrylate (CN) adhesives (Tokyo Measuring Instruments Lab Co., Ltd.). The RbV₃Sb₅ sample was glued on a sapphire substrate with ThreeBond 2086M two-component epoxy resin adhesive (ThreeBond Holdings Co., Ltd.). A schematic setup is shown in Fig. 1(a). Strain gauges (Tokyo Measuring Instruments Lab Co., Ltd.) were used to characterize the compressive (tensile) strain induced on MoTe₂ (RbV₃Sb₅) at cryogenic temperatures. The biaxial strain induced on MoTe₂ and RbV₃Sb₅ was found to be -1.06% and $+1.16\%$, respectively, with the positive sign indicating tensile strain as a convention. See the supplementary material for more details. The temperature-dependent resistance was measured by the standard four-probe method. All low-field data were collected with a Lakeshore AC resistance bridge in a Bluefors dilution refrigerator down to 10 mK. All high-field data were collected with a customized lock-in detection unit in a helium-3 cryostat with a pulsed field magnet at the National High Magnetic Field Laboratory's Pulsed Field Facility at Los Alamos National Laboratory.

To demonstrate the versatility of the biaxial strain methodology, we have applied compressive and tensile strain to MoTe₂ and RbV₃Sb₅, respectively. Both MoTe₂ and RbV₃Sb₅ are novel superconductors currently being investigated intensively.^{18–33} Figure 1(a) shows the schematic of the biaxial strain device. To prepare this device, a thin flake cleaved from a bulk single crystal is mechanically coupled to a piece of insulating substrate, which is ~ 10 -mm-thick, with a suitable adhesive at room temperature. As the device is cooled toward 0 K, the differential thermal expansion causes the substrate to exert biaxial strain on the thin flake sample. For the case of MoTe₂ on polycarbonate, the resultant strain at the lowest attainable

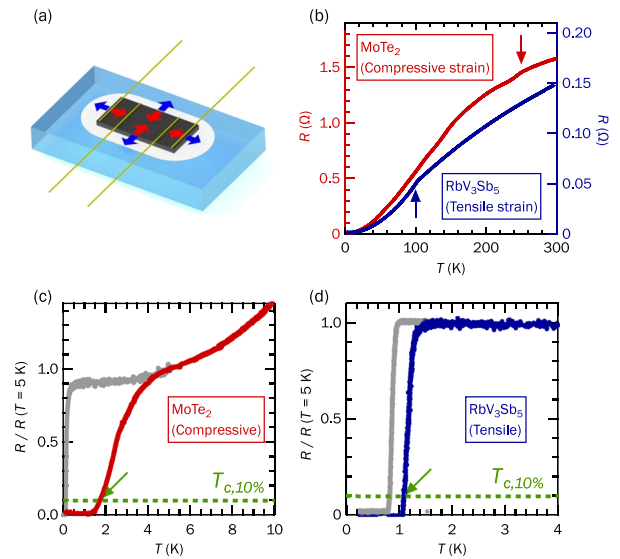


FIG. 1. (a) Schematics of the device for biaxially straining a sample (black) by utilizing the differential thermal expansion between the sample and the substrate (light blue). The sample is mechanically coupled to the substrate with a suitable adhesive (white). The device is equipped with four gold wires for magnetotransport studies. Both the compressive and tensile strains can be realized by a careful selection of the substrate. (b) Temperature dependence of the electrical resistance $[R(T)]$ of strained MoTe₂ (red) and RbV₃Sb₅ (blue) measured with the simple device architecture in Fig. 1(a), indicating high signal quality. The anomaly at $T_s \approx 250$ K for MoTe₂ (red arrow) can be attributed to the $1T'$ to T_d structural phase transition. On the other hand, RbV₃Sb₅ exhibits a charge density wave (CDW) transition at $T_{CDW} \approx 100$ K. These phase transition temperatures are consistent with the corresponding free-standing samples in previous studies.^{17,18,21,26,34,35} This is expected since the induced biaxial strain only fully develops at the 0 K limit and is not effective at elevated temperatures. (c) Normalized $R(T)$ curves showing the superconducting transitions of free-standing MoTe₂ (gray) and MoTe₂ under a compressive strain (red). (d) Normalized $R(T)$ curves showing the superconducting transitions of free-standing RbV₃Sb₅ and RbV₃Sb₅ under a tensile strain (blue). The horizontal dashed green lines indicate the “10% criterion” for the definition of T_c .

temperature is compressive, while for RbV₃Sb₅ on sapphire, tensile strain is induced.

The gold wires attached to the thin flake enable electrical transport measurements of the strained samples. Figure 1(b) shows the temperature dependence of resistance, $R(T)$, of MoTe₂ (red) and RbV₃Sb₅ (blue) measured with the simple device architecture in Fig. 1(a), indicating high signal quality. The anomaly at $T_s \approx 250$ K for MoTe₂ (red arrow) can be attributed to the $1T'$ to T_d structural phase transition. On the other hand, RbV₃Sb₅ exhibits a charge density wave (CDW) transition at $T_{CDW} \approx 100$ K. These phase transition temperatures are consistent with the corresponding free-standing samples in previous studies.^{17,18,21,26,34,35} This is expected since the induced biaxial strain only fully develops at the 0 K limit and is not effective at elevated temperatures.

To explore the effect of the biaxial strain at low temperatures, we turn to another quantum state, namely superconductivity. Figures 1(c) and 1(d) compare the normalized $R(T)$ for the strained MoTe₂ (red) and strained RbV₃Sb₅ (blue) against their free-standing counterparts (gray curves). Benchmarking the superconducting critical temperature (T_c) with the “10% criterion,” where the resistance drops to 10% of the normal state resistance, the application of strain boosts the T_c from 0.13 to 1.7 K for MoTe₂, and from 0.82 to 1.1 K for RbV₃Sb₅. The large enhancement of superconductivity

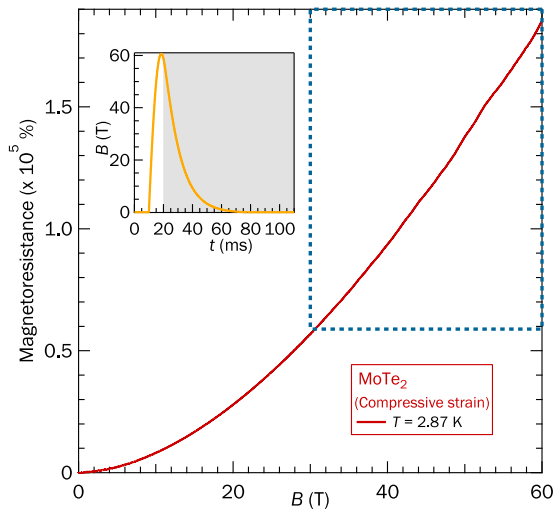


FIG. 2. Magnetoresistance of strained MoTe₂ at 2.87 K. The high field data in the dashed rectangle was used for SdH oscillation analysis. The top inset illustrates a representative magnetic-field pulse profile up to 60 T. The data from the descending field (shaded area) is used in the analysis. The field direction is perpendicular to the *ab*-plane of the sample.

in MoTe₂ due to biaxial compressive strain has also been pointed out recently in Ref. 23. Thus, the biaxial device is particularly powerful for exploring the ground state of layered quantum materials or for probing them near 0 K, prompting the investigation of the electronic structure of MoTe₂ and RbV₃Sb₅ in higher fields using a pulsed magnet.

The main panel in Fig. 2 shows the magnetoresistance of the strained MoTe₂ device measured up to 60 T at 2.87 K. The field dependence of the resistance has been corrected with a field-independent offset, as detailed in the supplementary material. The magnetic field was applied perpendicular to the *ab*-plane of the sample. A typical magnetic-field pulse profile is shown in the inset, and all data analyzed in this manuscript are from the portion collected with the descending field, indicated by the shaded region. The magnetoresistance curve in the main panel shows a nearly B^2 behavior, and the resistance is non-saturating even at 60 T. A non-saturating, nearly quadratic-in-field magnetoresistance has also been observed in the free-standing MoTe₂ as well as our previous strained sample.^{23,34,36} At 60 T, the magnitude of the magnetoresistance (MR) reaches ~185 000%. For $B \geq 30$ T, the magnetoresistance exhibits ripples (data enclosed in the dashed rectangle), which are, in fact, SdH oscillations.

After removing the large magnetoresistance background, clear SdH oscillations in strained MoTe₂ are immediately visible, with representative curves displayed in Fig. 3(a). The SdH oscillations deteriorate with increasing temperature, as expected. The fast Fourier transform (FFT) was performed between $B = 30$ T and $B = 60$ T for all temperatures to quantify these oscillations. The FFT spectra at different temperatures are plotted in Fig. 3(b). Three SdH frequencies, F_α , F_δ , and F_ω , are well resolved at 263, 800, and 1103 T, respectively. F_α is consistent with an electron pocket observed in previous studies on free-standing samples^{20,21,34,37} and also on a

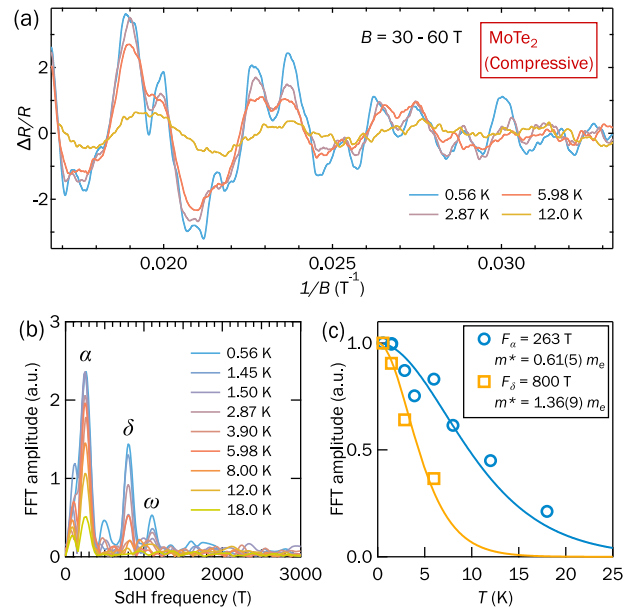


FIG. 3. (a) Representative SdH oscillations of strained MoTe₂ at different temperatures after removing the magnetoresistance background. (b) FFT spectra at various temperatures ranging from 0.56 to 18 K, revealing clear peaks at $\alpha \sim 263$ T, $\delta \sim 800$ T, and $\omega \sim 1103$ T. (c) Temperature dependence of the SdH amplitudes for F_α (open circles) and F_δ (open squares). The solid curves are the fittings by the Lifshitz-Kosevich (LK) formula [Eq. (1)] and extracted effective masses are shown.

biaxially strained MoTe₂.²³ F_δ , on the contrary, is likely related to the SdH frequency of 758 T reported for the free-standing MoTe₂.³⁴ In addition, F_ω is close to the sum of F_α and F_δ . To extract more information about these Fermi pockets, we analyze the SdH oscillations based on the thermal damping factor R_T of the Lifshitz-Kosevich (LK) theory, which can be expressed as

$$R_T = \frac{14.693m^*T/B}{\sinh(14.693m^*T/B)}. \quad (1)$$

From Eq. (1), we can extract the effective mass of the quasi-particles (m^*) on the extremal cyclotron orbits of the Fermi sheets perpendicular to the magnetic field direction. The effective masses of α and δ are found to be 0.61(5) and 1.36(9) times the bare electron mass m_e , respectively. The effective mass of ω cannot be extracted because the FFT amplitudes are too weak. The m^* of α is similar to that in the free-standing MoTe₂.^{20,34,37} However, the m^* of δ is significantly different from what is observed in free-standing MoTe₂, where $m^* = 1.99(6)m_e$.^{20,34,37} The above observations reveal that the δ pocket is much more sensitive to biaxial strain tuning than the α pocket. Considering the drastic enhancement in T_c by biaxial strain tuning, it is quite likely that the δ pocket plays a major role in this effect.

Next, we switch to the tensile-strained RbV₃Sb₅ and investigate its electronic properties. Figure 4(a) displays the $R(B)$ data of strained RbV₃Sb₅ at 0.56 K. The $R(B)$ curve shows a monotonically increasing magnetoresistance that reaches ~2800% at 50 T, indicating reasonably high sample quality.^{18,31} The $R(B)$ background

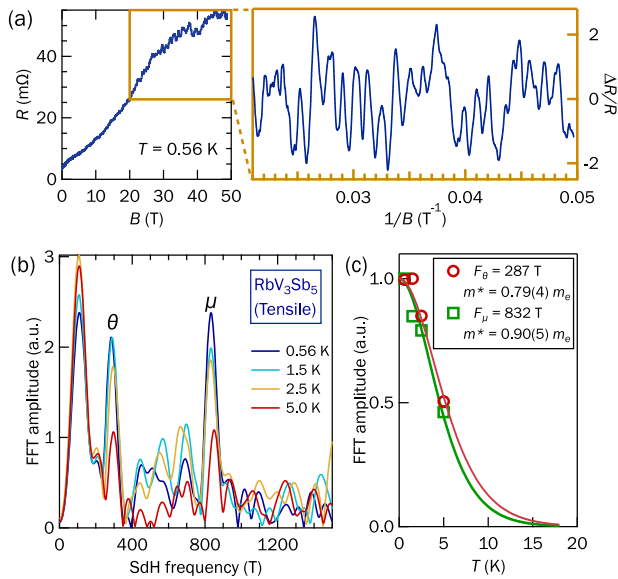


FIG. 4. (a) Magnetic field dependence of the electrical resistance for strained RbV_3Sb_5 at $T = 0.56$ K. A close-up view of SdH oscillation signals is shown on the right panel after background removal. (b) FFT spectra at various temperatures. (c) Temperature dependence of the SdH amplitudes for $F_\theta \sim 287$ T (open circles) and $F_\mu \sim 832$ T (open squares). The solid curves are the corresponding Lifshitz–Kosevich fittings [Eq. (1)], and the resultant effective masses are shown in the figure.

is removed by using a polynomial from 20 to 50 T, and the SdH oscillations are extracted. The resultant oscillatory signals are clearly visible and shown in the right panel of Fig. 4(a). The SdH oscillations at various temperatures were analyzed by FFT, and the spectra are plotted in Fig. 4(b). Here, we focus on SdH frequencies above 200 T, where FFT analysis at this field window (20–50 T) works well. To resolve frequencies lower than 200 T, good SdH signals down to lower fields are necessary. Two SdH frequencies, labeled as F_θ and F_μ , were resolved at 287 and 832 T in the spectra, respectively. Since the FFT spectra reported earlier for free-standing samples are complicated,^{18,26,30} we proceed by performing the LK analysis of F_θ and F_μ to look for correspondence between them and the SdH frequencies in free-standing RbV_3Sb_5 .

The effective mass analysis of F_θ and F_μ suggests an intricate dependence on biaxial strain. Figure 4(c) shows the LK fittings on the SdH amplitudes of F_θ and F_μ using Eq. (1). Surprisingly, we find that the m^* values for F_θ and F_μ are $0.79(4)m_e$ and $0.90(5)m_e$, respectively, which are substantially larger than all m^* values reported in free-standing RbV_3Sb_5 .^{18,26,30,31} The enhanced m^* is likely attributable to increased charge fluctuations, resulting from the weakening of the CDW order when the ab -plane is enlarged by the biaxial strain. Comparisons between uniaxial strain and hydrostatic pressure effects on the sister compound CsV_3Sb_5 have established that the shortening of the out-of-plane lattice constant c is the primary parameter driving the initial increase in T_c and the decrease in T_{CDW} .^{38,39} The decrease in c in uniaxial experiments is due to the Poisson effect. Thus, a decrease in c can be similarly expected when the biaxial tensile strain acts on RbV_3Sb_5 .

Indeed, the observed enhancement of T_c under biaxial strain lends support to this hypothesis, and the increased charge fluctuations can be felt by the quasiparticles as the CDW phase in RbV_3Sb_5 is progressively suppressed, resulting in enhanced m^* values. However, other possible scenarios that can lead to an enhancement in T_c , such as the change in charge carrier density induced by biaxial strain, should also be noted. Further Hall effect and detailed SdH measurements would shed light on the role of biaxial strain on T_c and m^* in RbV_3Sb_5 .

Our pilot magnetotransport measurements on MoTe_2 and RbV_3Sb_5 up to 60 T clearly demonstrate the feasibility of probing the electronic structures of layered materials tuned by a biaxial strain. While we are mindful of the drawbacks that the strain only fully develops at the zero temperature limit, the methodology offers an elegant and simple way to perturb and study the ground state of quantum materials. The small footprint of the biaxial device further facilitates integration with other pulsed-field magnets, and the flexible wiring arrangement can enable other high-field experiments. For instance, we do not foresee any technical barriers to measuring the Hall effect—regardless of whether it is classical or quantum, normal or anomalous—up to 100 T while simultaneously tuning the system by biaxial strain.

Hydrostatic pressure has been extremely successful in material tuning. However, it is challenging to integrate pressure devices into a pulsed-field magnet. This is because pressure devices are usually bulky and complicated, requiring elaborate effort to prepare, and they contain many metallic parts. The present methodology of utilizing the thermal expansion mismatch of an insulating substrate circumvents all these issues. Finally, while hydrostatic pressure is always compressive, biaxial strain can be either compressive or tensile with a careful selection of the substrate, as has been demonstrated in this manuscript. Therefore, the use of biaxial strain as a tuning parameter offers the possibility of addressing unique challenges in materials research, and its integration with a pulsed magnetic field holds great promise for advancing our knowledge in the field of quantum materials.

To summarize, we have applied biaxial strain using a convenient method on superconductors MoTe_2 and RbV_3Sb_5 and conducted magnetotransport measurements in a pulsed magnetic field. T_c of MoTe_2 increases substantially from 0.13 to 1.7 K upon the application of compressive biaxial strain by polycarbonate substrate. On the other hand, a sapphire plate is used to apply tensile strain to RbV_3Sb_5 , and its T_c has been raised from 0.82 to 1.1 K. We have observed quantum oscillations in magnetoresistance, the SdH effect, in both devices. In compressive-strained MoTe_2 , one of the SdH frequencies, F_δ , has shown a decrease in effective mass m^* compared to the free-standing sample, which could be relevant to the observed T_c enhancement under strain. In tensile-strained RbV_3Sb_5 , we observe substantial increases in the m^* values associated with F_θ and F_μ , implying that biaxial strain is effective in tuning electron correlations in this Kagome system. This study illustrates that the combination of biaxial strain and pulsed magnetic field provides a powerful platform for tuning and investigating topological materials under unprecedented conditions.

The details for the determination of biaxial strain induced in MoTe_2 and RbV_3Sb_5 and for the corrections of magnetoresistance in Fig. 2 are described in the supplementary material.

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AUTHOR DECLARATIONS

Conflict of Interest

The authors have no conflicts to disclose.

Author Contributions

King Yau Yip and Lingfei Wang contributed equally to this work.

King Yau Yip: Data curation (equal); Formal analysis (lead); Methodology (lead); Validation (lead); Writing – original draft (equal). **Lingfei Wang:** Data curation (equal); Formal analysis (equal); Writing – original draft (equal). **Tsz Fung Poon:** Formal analysis (equal); Writing – original draft (equal). **Kai Ham Yu:** Formal analysis (supporting). **Siu Tung Lam:** Data curation (supporting); Resources (supporting). **Kwing To Lai:** Data curation (supporting); Formal analysis (supporting); Resources (supporting). **John Singleton:** Data curation (equal); Methodology (equal); Resources (equal). **Fedor F. Balakirev:** Data curation (equal); Formal analysis (supporting); Methodology (equal); Resources (equal); Writing – original draft (supporting). **Swee K. Goh:** Conceptualization (lead); Formal analysis (equal); Funding acquisition (lead); Project administration (lead); Resources (lead); Supervision (lead); Validation (lead); Writing – original draft (equal).

DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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