

Pressure suppresses the density wave order in kagome metal LuNb_6Sn_6

William R. Meier^{1,*}, David E. Graf², Brenden R. Ortiz^{3,†}, Shirin Mozaffari¹, and David Mandrus^{1,4,‡}

¹Materials Science and Engineering Department, University of Tennessee Knoxville, Knoxville, Tennessee 37996, USA

²National High Magnetic Field Laboratory, Tallahassee, Florida 32310, USA

³Materials Science and Technology Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, USA

⁴Department of Physics and Astronomy, University of Tennessee Knoxville, Knoxville, Tennessee 37996, USA



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The density waves that develop in kagome metals ScV_6Sn_6 and LuNb_6Sn_6 at low temperature appear to arise from underfilled atomic columns within a V-Sn or Nb-Sn scaffolding. Compressing this network with applied pressure in ScV_6Sn_6 suppressed the structural transition temperature by constraining atomic rattling and inhibiting the shifts that define the structural modulation. We predicted that the density wave transition in LuNb_6Sn_6 at 68 K would be suppressed by pressure as well. In this Letter, we examine the pressure dependence of the density wave transition by measuring resistance vs temperature up to 2.26 GPa. We found the transition temperature is smoothly depressed and disappears around 1.9 GPa. This result not only addresses our prediction, but strengthens the rattling chains origin of structural instabilities in the HfFe_6Ge_6 -type kagome metals.

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Introduction. Researchers have been enamored by the kagome lattice for the last decades due to the exciting behaviors that can arise in materials hosting this structural motif. In metallic systems, the connectivity of atoms on kagome sites generates electronic structures with desirable states like flat bands, van Hove singularities, and Dirac nodes [1,2]. Among metallic kagome compounds, structural modulations have emerged as a reoccurring theme. The CsV_3Sb_5 family [3,4], CsCr_3Sb_5 [5], V_3Sb_2 [6], $\text{La}(\text{Ru},\text{Fe})_3\text{Si}_2$ [7], $\text{Yb}_{0.5}\text{Co}_3\text{Ge}_3$ [8], hexagonal-FeGe [9–11], and ScV_6Sn_6 [12] all develop periodic structural distortions on cooling.

The density wave instability observed below 92 K in ScV_6Sn_6 has received significant attention due to some of its peculiar features. These include curious transport properties [13–16] competing density wave instabilities [17–23] and subtle band structure reconstruction [24–29]. The structural modulation that develops in ScV_6Sn_6 has characteristics that distinguish it from traditional charge density waves (CDWs). In particular, the transition temperature can be suppressed by compressing the lattice with pressure [30] and by filling the lattice by larger rare earth elements [31].

This is different from usual behavior expected for the lattice volume dependence for CDW systems. Expanding the lattice through isovalent doping and shrinking it with pressure are expected to have opposite impacts on the Fermi-surface details that underlie CDW formation. Therefore, these system often have common trends of transition temperature with lattice volume with both doping and pressure [Fig. 1(a)] [32–35].

In fact, there is direct evidence that the modulation in ScV_6Sn_6 does not arise from the Fermi-surface nesting com-

mon for CDWs [17–19,23,28,36,37]. This motivated the development of a rattling model to understand the structural instability in the material [31]. In short, HfFe_6Ge_6 -type AM_6X_6 compounds can have rattling instabilities whenever a small A atom sits in a sufficiently large M_6X_6 scaffolding. In ScV_6Sn_6 this extra space leads to the formation of Sn-Sn bond modulation that characterize the low-temperature structure [18,37].

We recently demonstrated that this rattling instability model extends to the extensive LnNb_6Sn_6 family of compounds [38]. In the compounds with the smallest rare earths, x-ray experiments reveal rattling of rare-earth-Sn chains within the Nb-Sn network [Fig. 1(b)]. Crucially, LuNb_6Sn_6 develops a density wave (DW) order at approximately 68 K with static Sn-Sn bond modulation [Fig. 1(d)] analogous to that in ScV_6Sn_6 . We demonstrated that replacing lutetium with larger rare-earth elements not only inhibits the rattling at high temperatures, but also prevents the development of a DW [Fig. 1(e)].

This observation reinforced the rattling chain model developed for ScV_6Sn_6 [31] and inspired an important prediction: We hypothesized that the density wave transition temperature of LuNb_6Sn_6 will be suppressed with pressure [Fig. 1(a)]. We anticipate that rattling and Sn-Sn bond modulation will be inhibited as the Nb-Sn framework is compressed as seen in ScV_6Sn_6 [Fig. 1(c)].

In this Letter we present an expedient test of this idea by measuring the temperature-dependent resistance of LuNb_6Sn_6 under pressure to 2.26 GPa. We clearly observe that the transition temperature is suppressed as pressure is increased. The density wave in LuNb_6Sn_6 is suppressed by applied pressure, confirming our prediction. We further compare the pressure evolution of the density wave transition in ScV_6Sn_6 and Lu-doped ScV_6Sn_6 reported previously. Our work reaffirms the applicability of the rattling chain model in underfilled network compounds such as the HfFe_6Ge_6 -type materials.

*Contact author: javamocham@gmail.com

†Contact author: ortizbr@ornl.gov

‡Contact author: dmandrus@utk.edu

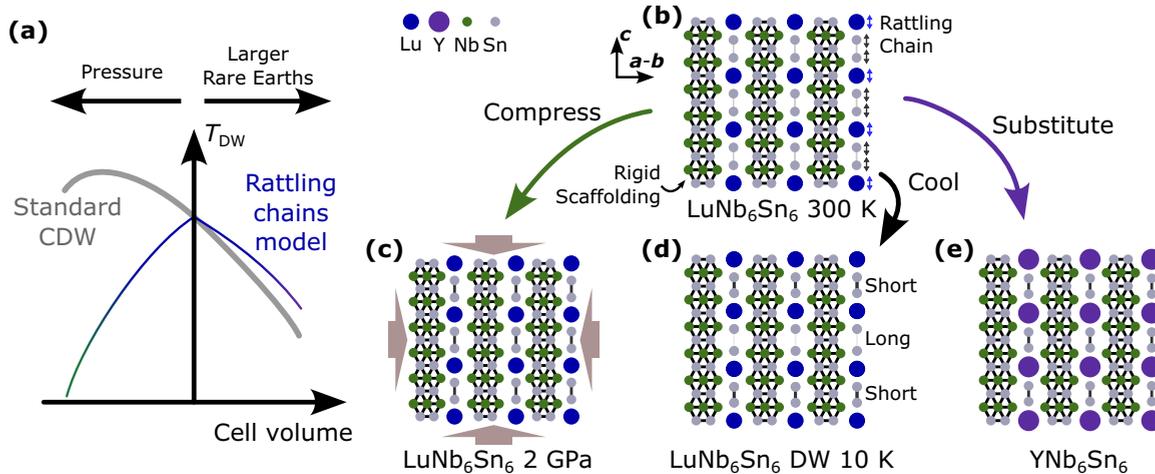


FIG. 1. (a) The transition temperature of traditional charge density wave (CDW) systems is generally expected to evolve smoothly with unit cell volume [32–35]. In contrast, density wave (DW) orders that develop due to underfilled rattling chains such as those in ScV_6Sn_6 and LuNb_6Sn_6 are expected to be suppressed by lattice expansion, by rare-earth doping, and by compression under pressure. (b) Cartoon structure of LuNb_6Sn_6 along the (110) plane. A rigid scaffolding of Nb and Sn atoms (green and gray) surround underfilled channels hosting a Lu-Sn-Sn chains. The extra space provided by the small Lu atoms allows the chains to rattle and displace, forming a DW structural modulation at low temperature (d). Applying pressure (c) and swapping Lu with larger rare earths (e) inhibits both rattling and DW formation.

Methods. Single-crystal growth of LuNb_6Sn_6 was performed using the self-flux technique [38]. Elemental reagents were combined in a ratio of 8:2:90 = Lu:Nb:Sn. We utilized Ames Laboratory Lu metal, Nb powder (Alfa, 99.9%), and Sn shot (Alfa, 99.9%). Reagents were placed into 5-mL Al_2O_3 (Canfield) crucibles fitted with a catch crucible/porous frit and sealed within fused quartz ampoules with approximately 0.6–0.7 atm of argon cover gas [39]. Samples were heated to 1150 °C and thermalized for 18 h before cooling to 900 °C at a rate of 1 °C/h. Samples are subsequently centrifuged at 900 °C. Single crystals are small, well-faceted, gray-metallic, hexagonal plates and blocks. The samples are stable in air, water, and common solvents.

For these high-pressure measurements, a 4-mm-i.d. hybrid cell was used with a BeCu outer wall and NiCrAl alloy for an insert. Copper wires and a fiber optic were fixed in place through the sample platform with Stycast 2850 FT epoxy. A hexagonal block-shaped crystal of LuNb_6Sn_6 was hand polished into a minuscule bar with a $80 \times 120 \mu\text{m}$ cross section using sandpaper. Contacts to the sample were made with 0.0005 in platinum wire in a conventional four-contact arrangement held in place with Epotek H20E silver epoxy cured for 30 min at 130 °C. The pressure medium is Daphne 7575 oil which has been shown to remain hydrostatic to pressures above 3 GPa [40]. Transport measurements of the sample were made with a Lake Shore 372 resistance bridge with 3708 pre-amp/scanner using 1 mA and a 5 s averaging time. All data were collected with cooling curves taken at a rate of 0.4 K/min. Accurate temperatures were ensured by fixing a Cernox thermometer to the outside of the pressure cell and allowing for additional exchange gas in the sample space of the physical properties measurement system (PPMS) where the experiment took place. Resistance data were averaged over 15 measurements to reduce the noise in the derivative. Raw data are provided in the Supplemental Material [41].

The fiber optics ends were cleaved cleanly to allow for a small chip of ruby to be glued to the end inside of the sample volume. Green laser light (532 nm) was transmitted to the ruby chip and the return fluorescence spectrum was determined by an Ocean Optics spectrometer. Ruby peaks are known to shift at a rate of 0.365 nm/GPa [42] and a room-temperature and low-temperature value for pressure were recorded for each curve. The reported pressures are from the region of interest (low temperatures) where the charge density wave was suppressed.

Results. Figure 2(a) presents the normalized resistance versus temperature data obtained for a series of pressure. The labeled pressures were estimated at the base temperature. At the lowest pressure, resistivity drops sharply on cooling through 64 K. This corresponds to the development of the density wave modulation observed by x-ray diffraction at zero pressure at 68 K [38]. In ScV_6Sn_6 this drop has been tied to an increase in carrier mobility in the density wave phase [12,43].

The shape of the resistance curve evolves neatly with increasing pressure (lighter colors). The clifflike drop at 0.1 GPa evolves into a cusp-shaped peak at 0.54 and 0.78 GPa. This further morphs into a broader step-up on cooling exemplified by the 1.26 GPa curve. The pressure evolution of this resistance feature mimics that observed in ScV_6Sn_6 [30] and $(\text{Sc}_{0.944}\text{Lu}_{0.054})\text{V}_6\text{Sn}_6$ [31]. Importantly, the high-pressure jump-up feature is most dramatic in LuNb_6Sn_6 .

In Fig. 2(a) there is a sharp drop in resistance between 2.8 and 3.7 K. A linear fit to the onset temperature versus pressure yields a slope of -0.41 ± 0.01 K/GPa with an intercept of 3.74 ± 0.02 K. [This agrees exquisitely with the pressure dependence the superconducting critical temperature of tin metal $T_c = 3.733 - (0.43 \text{ K/GPa})p$] [44]. Despite careful polishing, there is a Sn inclusion in our sample.

The evolving nature of the density wave transition feature in this dataset necessitates a systematic approach to extracting

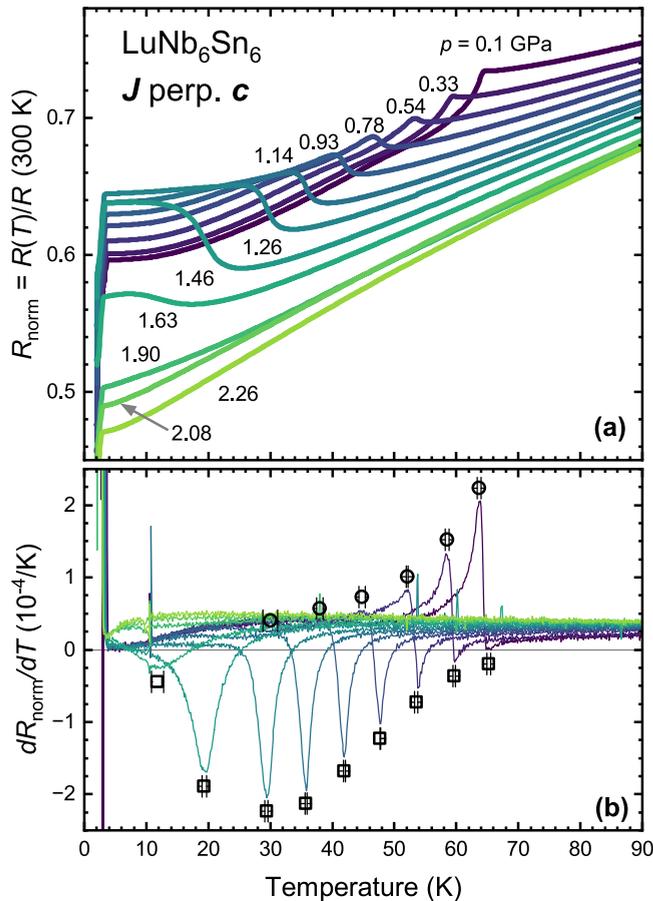


FIG. 2. (a) Normalized resistance vs temperature data for LuNb_6Sn_6 at increasing pressures measured on cooling. The sharp drops near 3 K are from a tin impurity. (b) Temperature derivatives of resistivity emphasize transition features. Maxima and Minima of the derivative are marked with circles and squares, respectively. Error bars communicate estimated temperature uncertainty.

the transition temperatures. We plot the temperature derivative of resistance presented in Fig. 2(b) to assist us. We selected the maxima and minima in $\frac{dR_{\text{norm}}}{dT}(T)$ as characteristic temperatures as they correspond to the positive and negative slope extremes of $R_{\text{norm}}(T)$. We identify these temperatures with estimated uncertainties with circles and squares in Fig. 2(b).

At low pressures where a step-down feature is observed, the maximum of $\frac{dR_{\text{norm}}}{dT}(T)$ is the best descriptor of this first-order transition temperature. In contrast, at higher pressure (such as 1.26 GPa) the steepest negative slope of $R_{\text{norm}}(T)$ appears to be the best descriptor for a broadened first-order transition. At intermediate pressures, we could imagine the transition temperature at the cusp in which naturally lies between the maxima and minima of $\frac{dR_{\text{norm}}}{dT}(T)$.

Importantly, we see no evidence of bulk superconductivity emerging as the density wave disappears. This was also true for ScV_6Sn_6 under pressure [30], doped ScV_6Sn_6 , and Lu-doped ScV_6Sn_6 under pressure [25,31]. The first-order nature of this transition in these compounds may provide a poor starting point for building a superconducting state.

Figure 3 presents the phase diagram of LuNb_6Sn_6 under pressure. The purple data clearly reveal that the transition

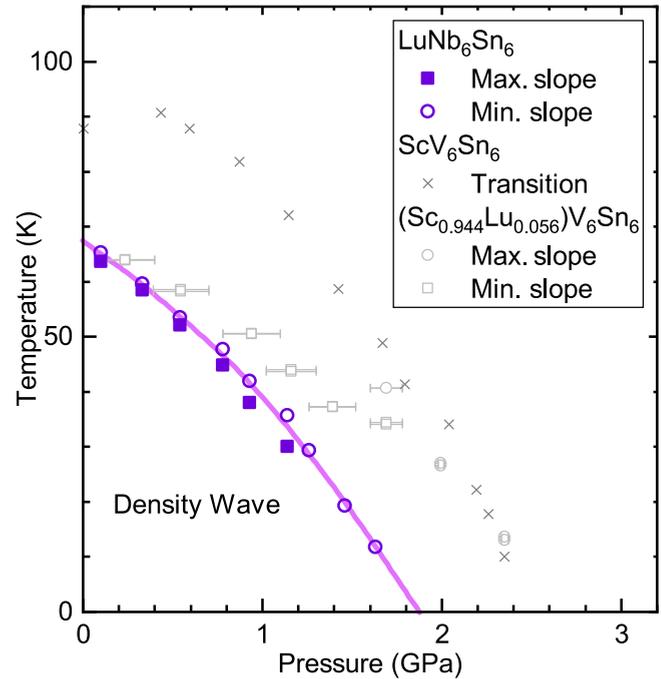


FIG. 3. Phase diagram reveals the suppression of the phase transition in LuNb_6Sn_6 with pressure. Transition temperatures from ScV_6Sn_6 [30] and Lu-doped ScV_6Sn_6 [31] are included, showing higher critical pressures.

temperature is smoothly suppressed with applied pressure and disappears by roughly 1.9 GPa. The solid line is included to emphasize the trend.

Discussion. We have confirmed our prediction that the density wave phase in LuNb_6Sn_6 would be suppressed by pressure. This supports the rattling chains picture developed for ScV_6Sn_6 [31] we extended to LuNb_6Sn_6 [38]. Compressing the Nb-Sn scaffolding removes the extra space in the structure, inhibiting DW formation.

It is also possible that compressing the lattice also modifies electronic structure in a way that could disfavor DW development. We suspect these electronic changes are less important based on numerous studies finding that Fermi-surface details are not the key driver of DW order in ScV_6Sn_6 [17–19,23,28,36,37]. To address this possibility, we propose studying Sc-substituted LuNb_6Sn_6 to further examine the importance of steric effects in DW development. The rattling chains picture would predict that replacing Lu with tiny Sc would exacerbate the underfilling of the scaffolding increasing the DW transitions temperature.

Demonstrating broader applicability of the rattling chain model beyond the $RV_6\text{Sn}_6$ family strengthens its value for materials design. Displacement fluctuations can be engineered by underfilling channels in compounds with rigid scaffolding networks. This could enhance thermoelectric performance and induce density wave transitions. In addition, identifying the critical pressure required to prevent density wave formation will aid computational efforts exploring the criteria for structural instabilities in the HfFe_6Ge_6 -type metals.

The pressure evolution of the transition temperatures in ScV_6Sn_6 and $(\text{Sc}_{0.944}\text{Lu}_{0.056})\text{V}_6\text{Sn}_6$ are plotted Fig. 3 [34]. Curiously, although the Lu-doped material has a significantly

reduced transition temperature at zero pressure, it shares a comparable critical pressure with ScV_6Sn_6 . LuNb_6Sn_6 has a notably lower critical pressure. This difference might be a result of the Nb-Sn framework being more compressible than the V-Sn. This should be addressable with DFT calculations or diffraction experiments under pressure.

Next we turn to a trend common to all three materials, the evolution of the shape of the resistance transition signature with pressure. In most density wave phases the resistance is higher in the low temperature phase due to a loss of carrier when band reconstruct [45]. In ScV_6Sn_6 , only the Sn1 p bands show significant reconstruction accounting for a minor reduction in carrier concentration [19,23,24,46]. It was proposed that the reduced resistance in the DW phase is due to sudden loss of strong scattering [12,13,43] by low-energy phonons associated with the rattling chains [17,18,20]. It is likely the same explanation lies behind the step-down transition feature observed in LuNb_6Sn_6 at zero pressure.

Why then does the transition step difference change sign on increasing pressure in both ScV_6Sn_6 and LuNb_6Sn_6 ? First, this might represent a crossover in the relative impact of the change of carrier concentration and mobility as the density wave develops. Alternatively, there may be a change in the nature of the density wave order at higher pressures. For example, applying pressure might favor the competing $1/3 \times 1/3 \times 1/2$ density wave [17,18,20] over the $1/3 \times 1/3 \times 1/3$ observed at zero pressure in both compounds [12,38]. These ideas could be tested by low-temperature Raman

spectroscopy, optical conductivity, or x-ray scattering under pressure.

Conclusion. The density wave instabilities of ScV_6Sn_6 and LuNb_6Sn_6 result from the extra space provided by undersized atoms in a rigid V-Sn or Nb-Sn network. In the scandium compound, compressing this network with pressure inhibits the displacements that constitute the density wave, reducing the transition temperature. We predicted that the transition should be suppressed with pressure in LuNb_6Sn_6 as well. Our temperature-dependent resistance measurements confirm this. This observation emphasizes the value of the rattling chain model for understanding the tunability of density wave instabilities in filled network compounds.

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Data availability. The data that support the findings of this article are openly available [47].

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